

## Note

**THEORETICAL STUDY OF THE KINETIC ANALYSIS OF THE DENSIFICATION OF CERAMIC POWDER COMPACTS BY “QUASI-ISOTHERMAL” DILATOMETRY**

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Paulik and Paulik [1,2] have developed a new thermodilatometric technique, designated “quasi-isothermal dilatometry” (QID), that allows the monitoring of the temperature of the sample in such a way that the measurements are performed at a constant rate of change in length. These authors have applied this technique to the study of the thermal decomposition of anhydrous  $\text{CaC}_2\text{O}_4$  and have concluded that QID allows discrimination of the two steps taking place in the thermal decomposition of this compound much better than the conventional thermodilatometry (TD) carried out at a constant heating rate.

It is noteworthy that the quasi-isothermal dilatometry has not been used to date for performing the kinetic analysis of shrinkage of compact powders. In the opinion of Sørensen [3] this technique is not useful in kinetic studies because the temperature increases continuously but not at a constant rate. Thus, Sørensen seems to understand that the drawback of QID is that, the temperature not being a known function of time, the densification of the compact cannot be represented as a simultaneous function of both temperature and time. A similar conclusion was reached by Paulik and co-workers [4] with regard to the kinetic analysis of thermogravimetric curves of the thermal decomposition of solids obtained at a constant decomposition rate. However, this interpretation does not agree with that of Criado et al. [5,6] who have successfully applied the thermogravimetry at constant decomposition rate to determine the decomposition mechanism of alkaline-earth carbonates.

The scope of the present paper is to develop a method that permits the kinetic analysis of thermodilatometric data obtained by means of the so called quasi-isothermal heating technique.

Sintering is an activated process that can be described [3,7] by the following general equation

$$\frac{\Delta l}{l_0} = (kt)^n \quad (1)$$

where  $\Delta l$  is the change in length of the specimen,  $l_0$  is the initial length,  $n$  is an exponent which depends on the sintering mechanism, and  $k$  is the rate

constant given by the Arrhenius law

$$k = A \exp(-E/RT) \quad (2)$$

where  $E$  and  $A$  are the activation energy and the pre-exponential factor of Arrhenius, respectively.

By differentiating eqn. (1) with respect to time  $t$ , we get

$$\frac{d(\Delta l/l_0)}{dt} = nk \left(\frac{\Delta l}{l_0}\right)^{1-(1/n)} \quad (3)$$

which, taking into account eqn. (2), becomes

$$\frac{d(\Delta l/l_0)}{dt} = nA \exp(-E/RT) \left(\frac{\Delta l}{l_0}\right)^{1-(1/n)} \quad (4)$$

If the thermodilatometric curve is recorded at a constant rate of change in length  $C = d(\Delta l/l_0)/dt$ , eqn. (4), after taking logarithms and rearranging, can be written as

$$\left(\frac{1}{n} - 1\right) \ln \frac{\Delta l}{l_0} = \ln \frac{nA}{C} - \frac{E}{RT} \quad (5)$$

or

$$\ln \frac{\Delta l}{l_0} = \left(\frac{n}{1-n}\right) \ln \frac{nA}{C} - \frac{E'}{RT} \quad (6)$$

where

$$E' = \left(\frac{n}{1-n}\right) E \quad (7)$$

The plot of the left hand side of eqn. (6) against  $1/T$  should be a straight line whose slope and intercept permit the calculation of the activation energy and the pre-exponential factor of Arrhenius, respectively.

The method of kinetic analysis described above has the inconvenience that it does not allow determination of the value of  $n$  and, therefore, without prior knowledge of this parameter, the actual value of the activation energy,  $E$ , cannot be calculated. However, this drawback can be overcome by recording a series of dilatometric curves at different constant rates of change in length. Hence, it is very easy to deduce from eqn. (5) that, at a constant value of  $(\Delta l/l_0)_c$ , the following relationship must be accomplished

$$\ln C = B - \frac{E}{RT_c} \quad (8)$$

where  $B = [\ln nA - (1/n - 1) \ln(\Delta l/l_0)_c]$  is a constant and  $T_c$  is the temperature at which the value of  $(\Delta l/l_0)_c$ , previously chosen, is reached on the thermodilatometric curve recorded at the constant rate of change in length  $C$ . Accordingly, the plot of  $\ln C$  vs.  $1/T_c$  should be a straight line whose slope gives the actual activation energy  $E$ .

The value of  $n$  can be calculated by comparing [by means of eqn. (7)] the  $E'$  value determined from eqn. (6) with that of  $E$  obtained from eqn. (8).

In summary, we can conclude that the densification mechanism of ceramic powder compacts can be characterized from the kinetic analysis of a set of quasi-isothermal dilatometric curves obtained at different constant rates of change in length.

#### REFERENCES

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